

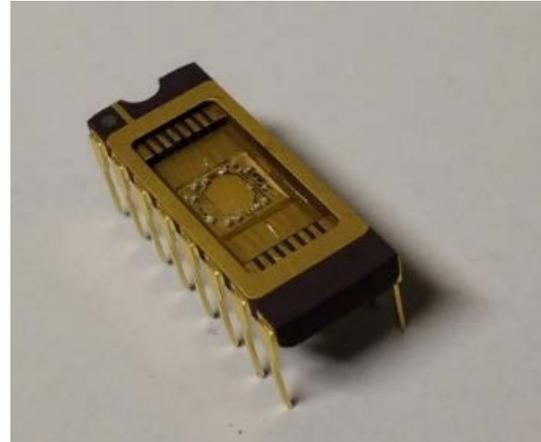
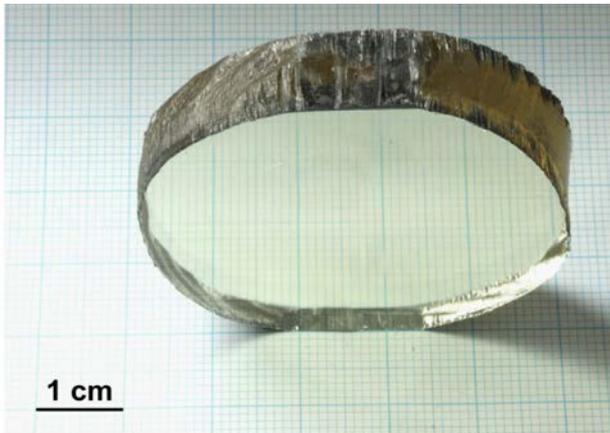


# Transport properties in $\beta$ -Ga<sub>2</sub>O<sub>3</sub> thin films

Saskia F. Fischer

## Novel Materials Group, HU Berlin

Johannes Boy, Robin Ahrling, Martin Handwerg, Rüdiger Mitdank



DFG

FI 932/10

FI 932/11



Leibniz Institute for Crystal Growth

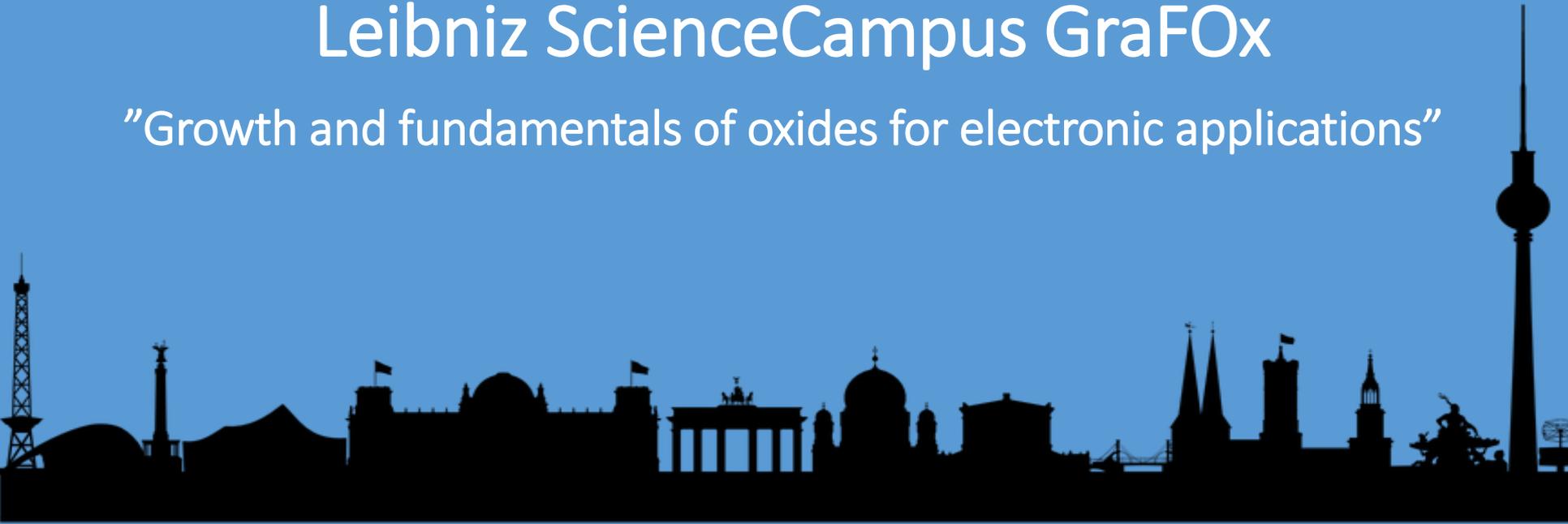


Andreas Popp/Günther Wagner,  
Zbigniew Galazka



# Leibniz ScienceCampus GraFOx

“Growth and fundamentals of oxides for electronic applications”



# A transparent wide-band gap semiconductor



Galazka *et al.*,  
*Journal of Crystal Growth* **404**, 184–191 (2014).

**Single crystal growth: Z. Galazka**

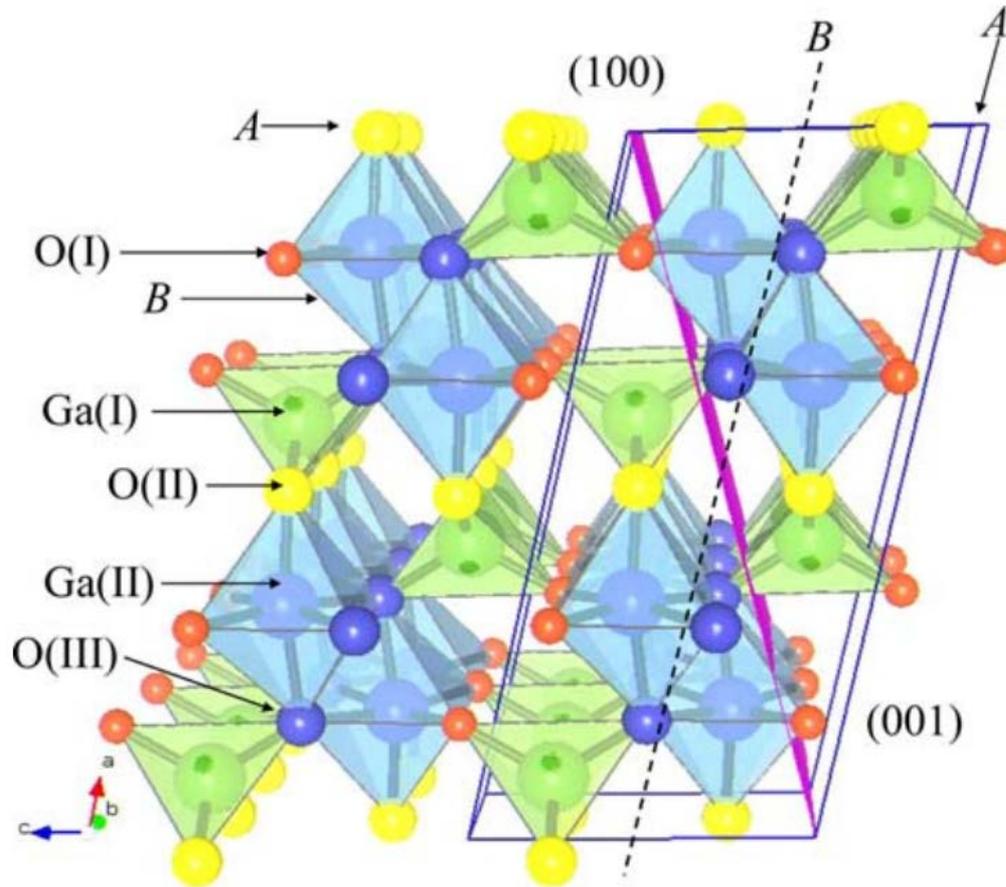
Leibniz Institute for Crystal Growth, Berlin, Germany



# $\beta$ -Ga<sub>2</sub>O<sub>3</sub> : Monoclinic crystal structure

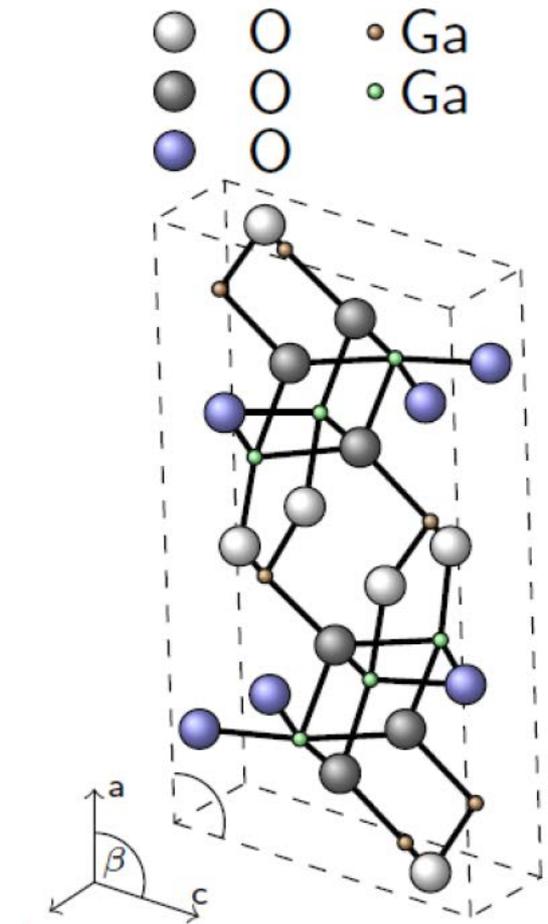
$$\begin{aligned} a &= 12.214 \text{ \AA} \\ b &= 3.037 \text{ \AA} \\ c &= 5.798 \text{ \AA} \end{aligned}$$

$$\beta = 103.83^\circ$$



cleavage planes:

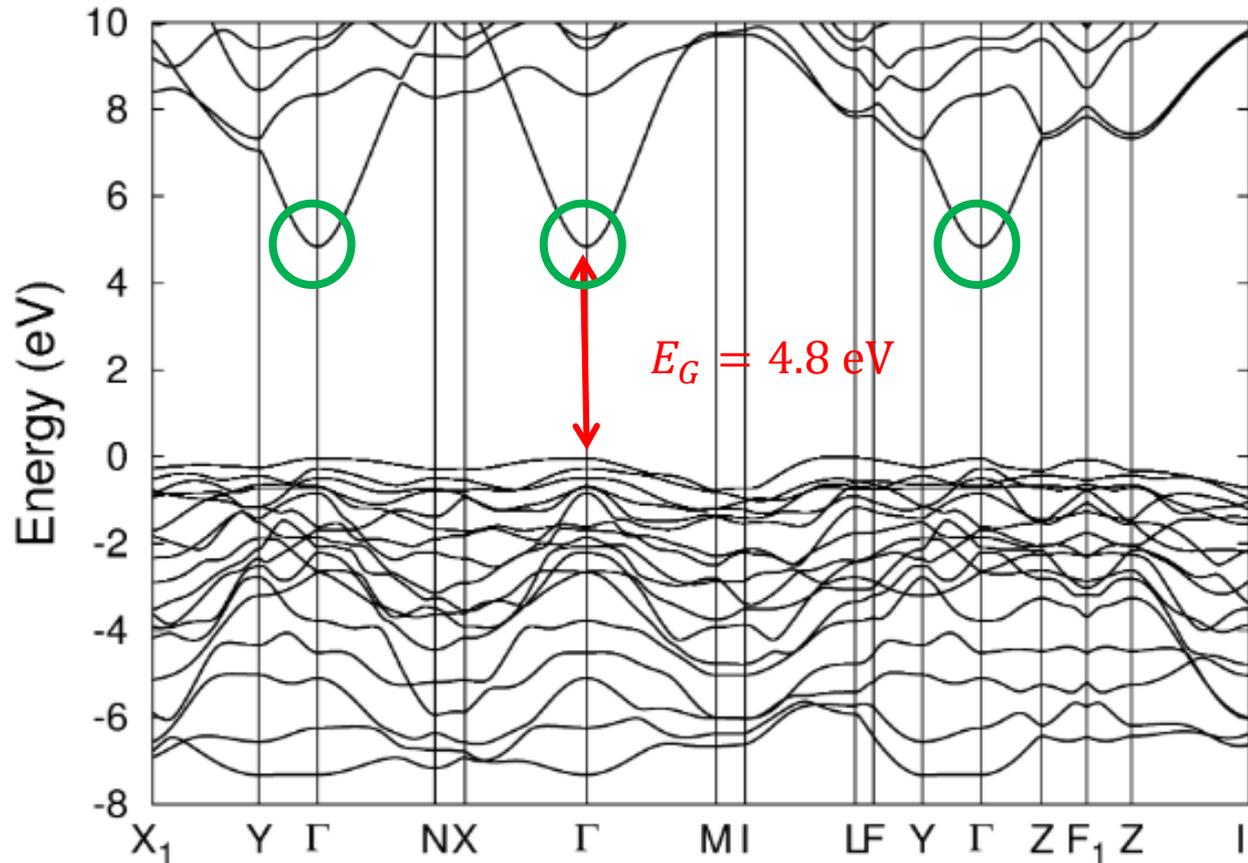
(100), (001)



# $\beta$ -Ga<sub>2</sub>O<sub>3</sub> : Electronic band structure

Charge carriers:  
Electrons

Transport:  
parabolic  
band approx.



effective  
electron  
mass

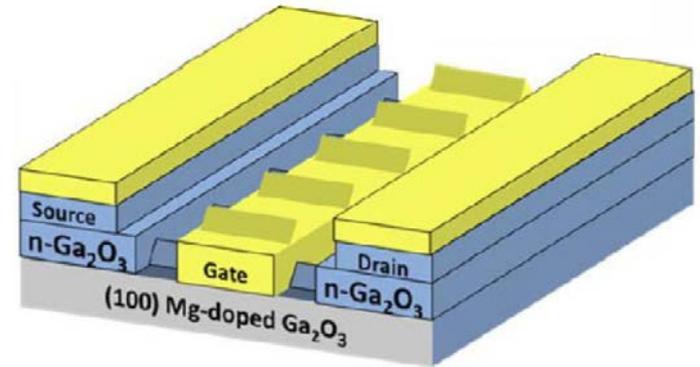
$0.313 m_e$

Mohamed, *et al.*,  
*Journal of Physics:  
Conference Series*  
**286**, 012027  
(2011).

# Opportunity of high breakdown field in Ga<sub>2</sub>O<sub>3</sub>

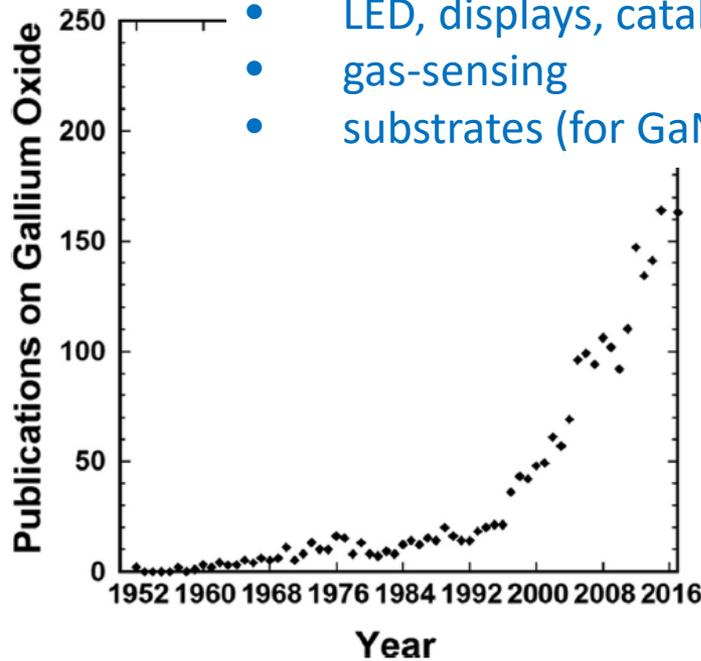
for high-power-devices

- $E_{br}$  predicted to be  $\sim 8$  MV/cm
- key advantage of Ga<sub>2</sub>O<sub>3</sub>
- Larger than the theoretical limits for GaN and SiC



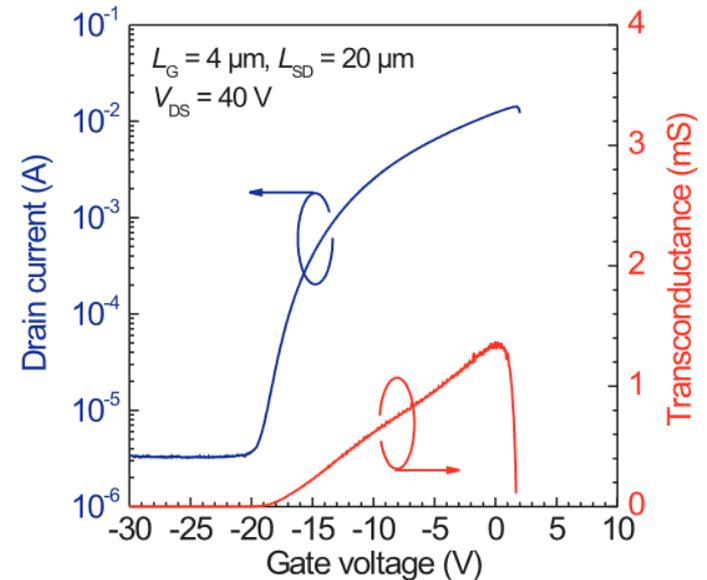
applications also in...

- LED, displays, catalysts
- gas-sensing
- substrates (for GaN)



Pearnton *et al.*, *Appl. Phys. Rev.* **5**, 011301 (2018).

Chabak *et al.*,  
*Appl. Phys. Lett.* **109**, 213501 (2016).



M. Higashiwaki, *et al.*;  
*Appl. Phys. Lett.* **100**, 013504 (2012).  
Review (2022)

# Material properties

Material Parameter	Si	GaAs	4H-SiC	GaN	Diamant	$\beta$ -Ga <sub>2</sub> O <sub>3</sub>	ZnGa <sub>2</sub> O <sub>4</sub>
Bandlücke $E_g$ [eV]	1.14	1.43	3.25	3.4	5.5	4.8	4.6
Dielektrizitätskonstante $\epsilon_s$	12	13	10	9	5.5	11	9.9
Durchbruchfeld $E_{Cr}$ [MV/cm]	0.3	0.4	2.5	3.3	10	8	6.5
Elektronenbeweglichkeit $\mu$ [cm <sup>2</sup> /(Vs)]	1450	8400	1000	1200	2000	300	107
Wärmeleitfähigkeit $\lambda$ [W/(mK)]	150	50	370	250	2000	10-30	22

## Bulk single crystals:

### Czochralski-growth

- As-grown, no doping
- $n = 9 \cdot 10^{16} - 6.5 \cdot 10^{17} \text{ cm}^{-3}$
- $d = 233 - 525 \mu\text{m}$



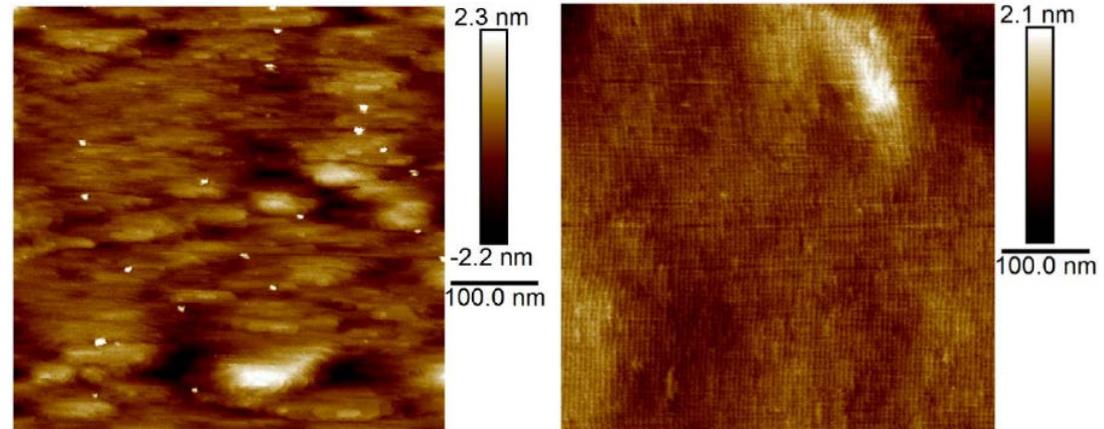
Galazka *et al.*,  
*J. Crystal Growth* **404**,  
184–191 (2014).

Z. Galazka, IKZ

## Epitaxial layers:

### Metallorganic Vapor Dep. (MOCVD)

- Si-Doping
- $n = 2.5 \cdot 10^{17} - 1.6 \cdot 10^{18} \text{ cm}^{-3}$
- $d = 25 - 225 \text{ nm}$
- 2D island growth & (rough) **step flow growth**



Mohamed, *et al.*, *J. Physics: Conf. Series* **286**, 012027 (2011).

Groups of G. Wagner / A. Popp, IKZ

## $\beta$ -Ga<sub>2</sub>O<sub>3</sub> – A transparent wide-band gap *semiconductor*

- **Electrical conductivity**
- **Thermal conductivity**
- **Thermo-electricity**

bulk  
&  
thin film effects

**Outlook: Giant-phonon drag *increase* by thin-film design**

# Thermo-/electric micro measurement platform for bulk & thin films

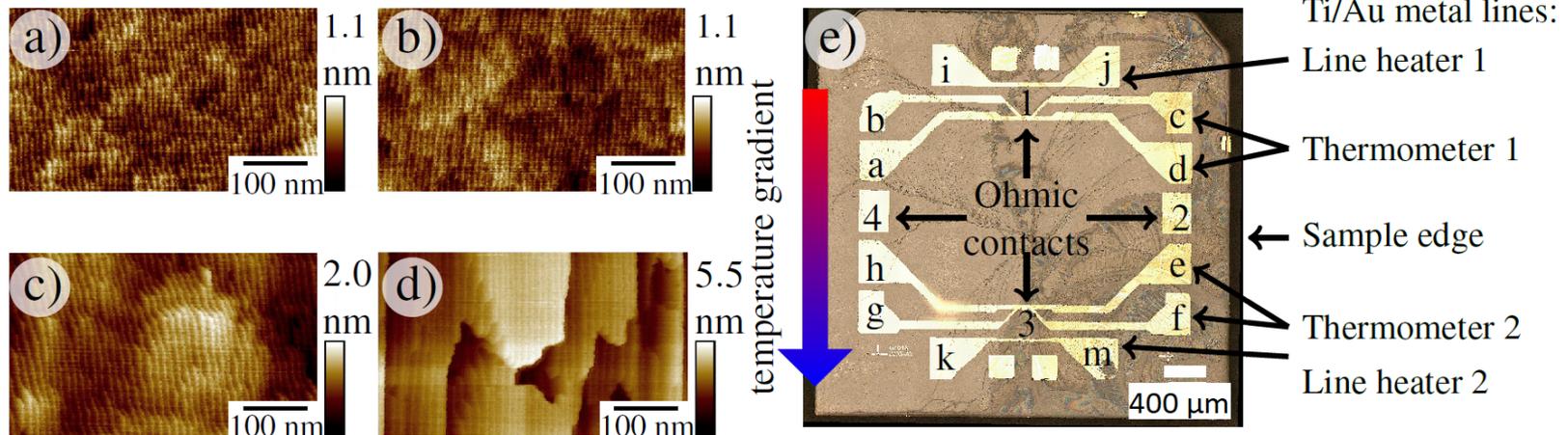
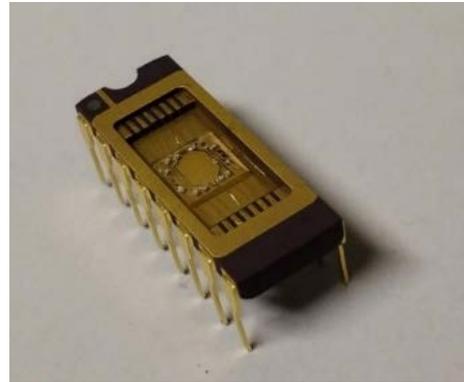
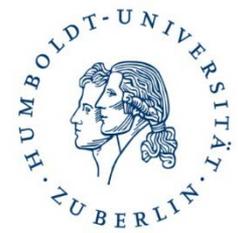


FIG. 1. (a) to (d) exemplary AFM measurement results of the investigated samples. (a) is the substrate of the  $d = 152$  nm sample, (b) the substrate of the  $d = 50$  nm sample, (c) the  $d = 152$  nm thin film and (d) the  $d = 50$  nm thin film. (e) Microscopic picture of a  $\beta$ - $\text{Ga}_2\text{O}_3$  thin film with the thermoelectric measurement platform consisting of Ti/Au (7 nm/35 nm) metal lines. Ohmic contacts were achieved by Al-wedge bonding at (1) - (4). The line heater and thermometer (a) - (m) were contacted by gold wires with indium contacts.

# Electrical properties

**Bulk**

**Thick epitaxial films**

**Thin epitaxial films**



Limiting effects of electron mobility  
- even vor ideal thin films

# A „zoo" of scattering mechanisms

phonon   
electron 

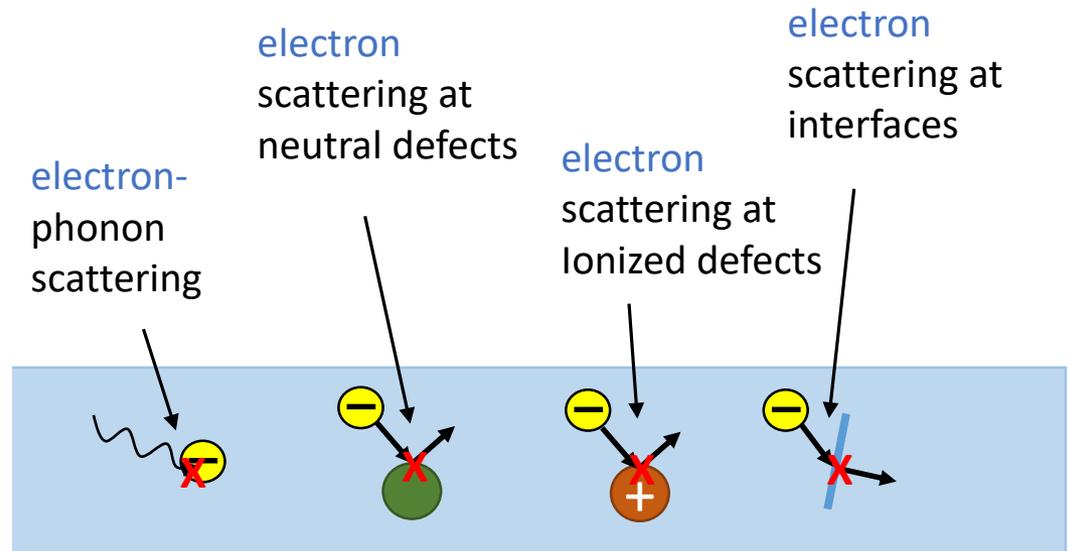
$$\sigma = \frac{e^2 n}{m^*} \langle \tau \rangle, \quad R_H = \frac{1}{ne} \frac{\langle \tau^2 \rangle}{\langle \tau \rangle^2}$$

## Reduced by...

Temperature

Expitaxy

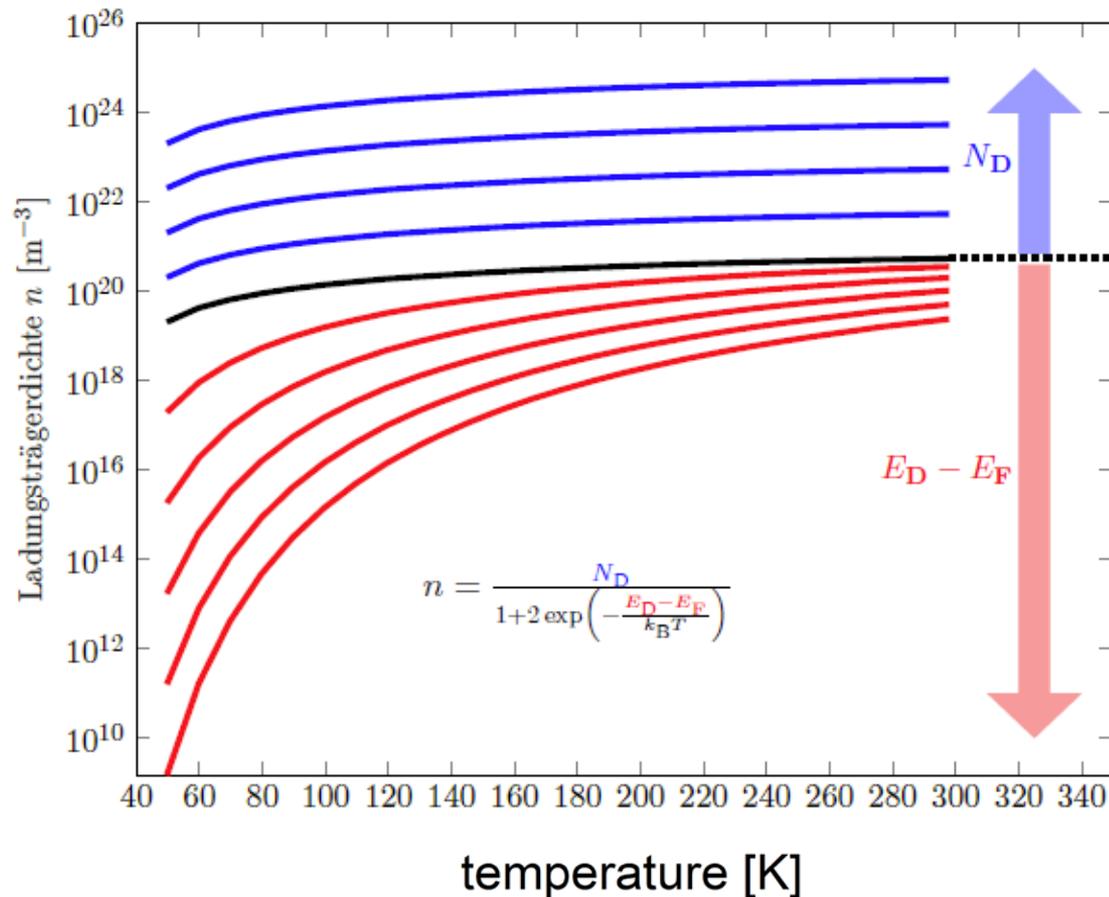
Homoepitaxy



# Electrical properties

## Electron density

$$n = \int_0^{\infty} D(E) f(T, E) dE \quad n = \int_0^{\infty} \frac{(2m^*)^{\frac{3}{2}}}{2\pi^2 \hbar^3} \cdot \frac{\sqrt{E + E_F - E_C}}{1 + \exp\left(\frac{E}{k_B T}\right)} dE$$

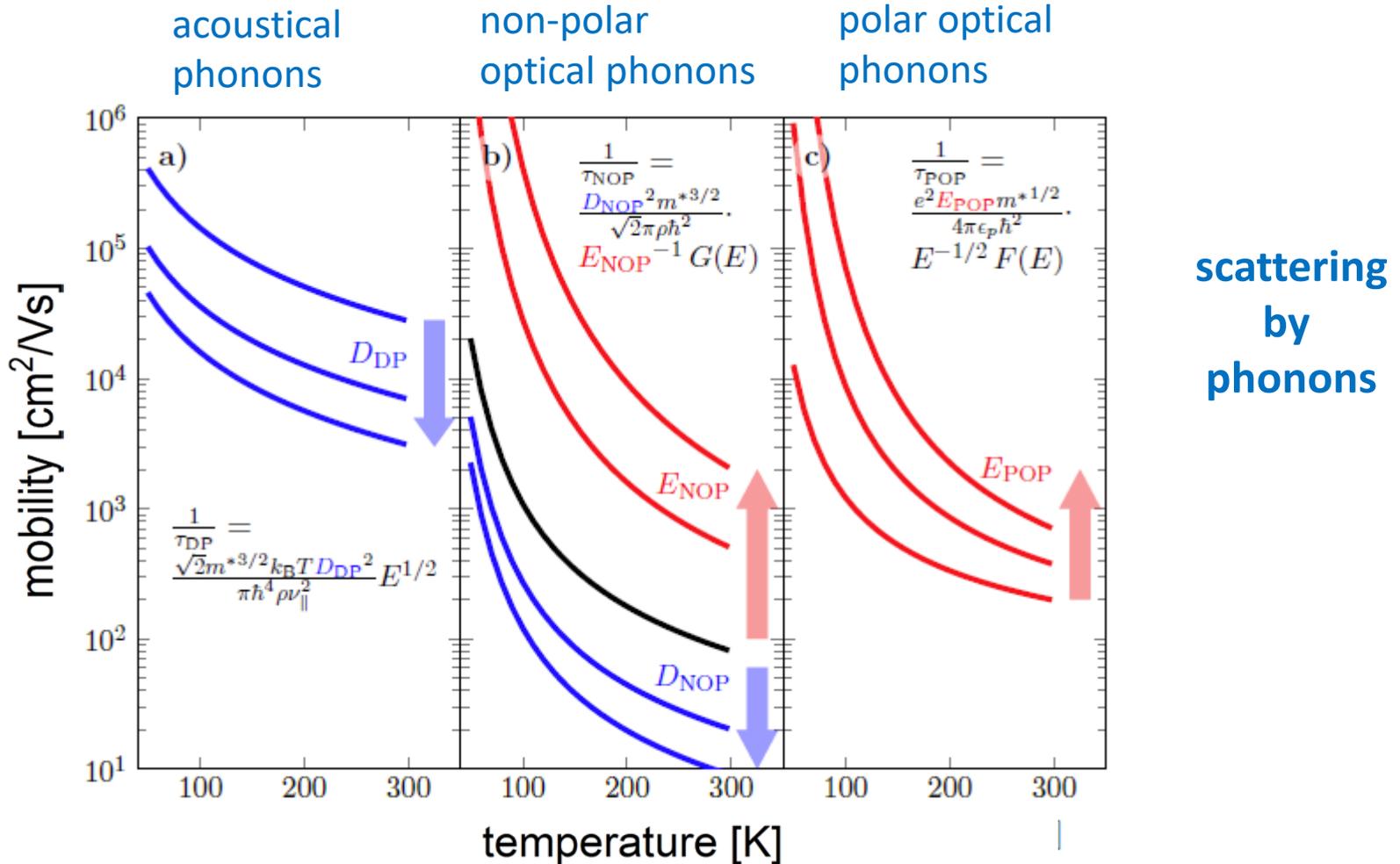


T. Oishi, *et al.*;  
Appl. Phys. Express **8**, 031101 (2015).

# Electrical properties

Electron mobility

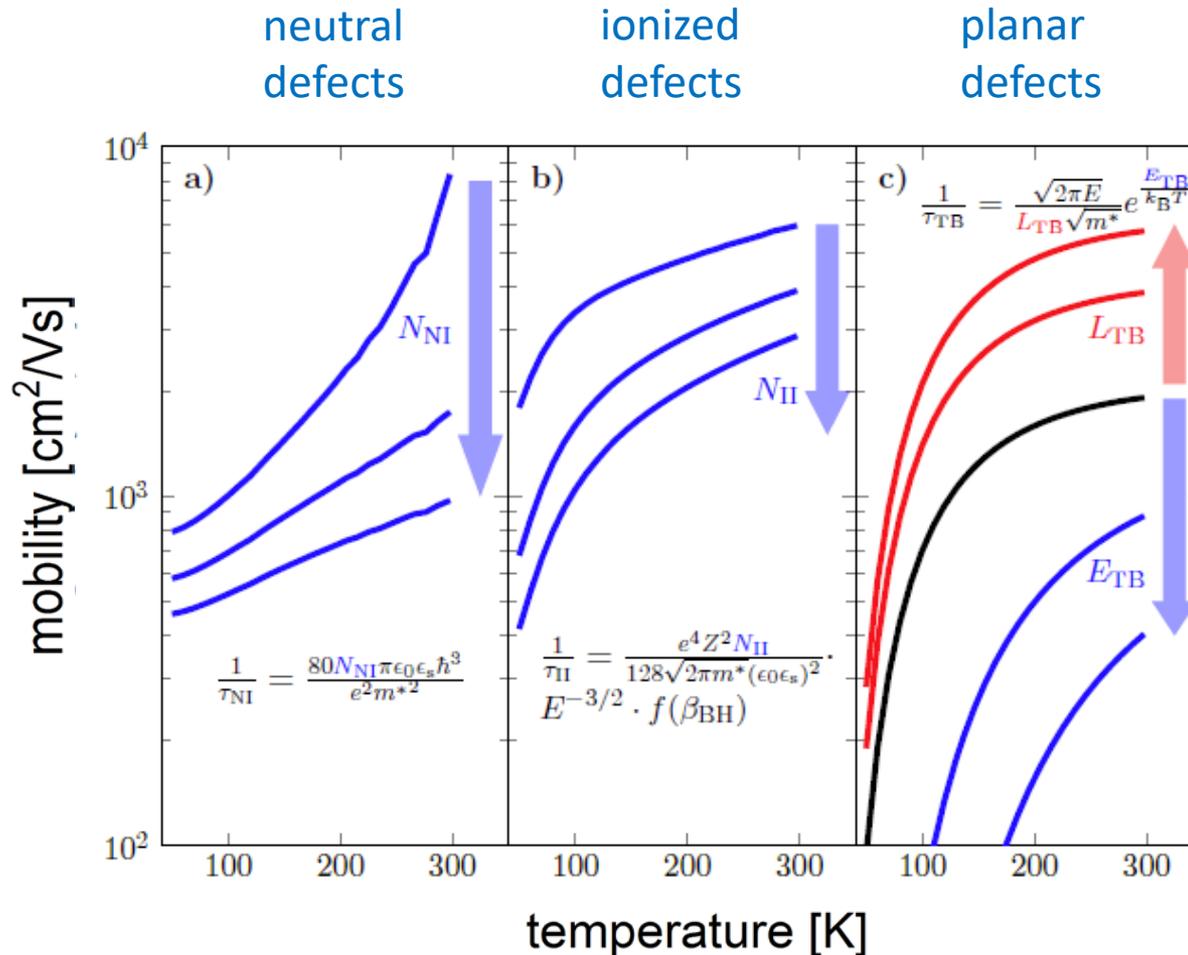
$$\mu = \frac{e\langle\tau_m\rangle}{m^*} = \frac{e}{m^*} \frac{\int_0^\infty E^{3/2} \tau_m(E) f(E) dE}{\int_0^\infty E^{3/2} f(E) dE}.$$



# Electrical properties

Electron mobility

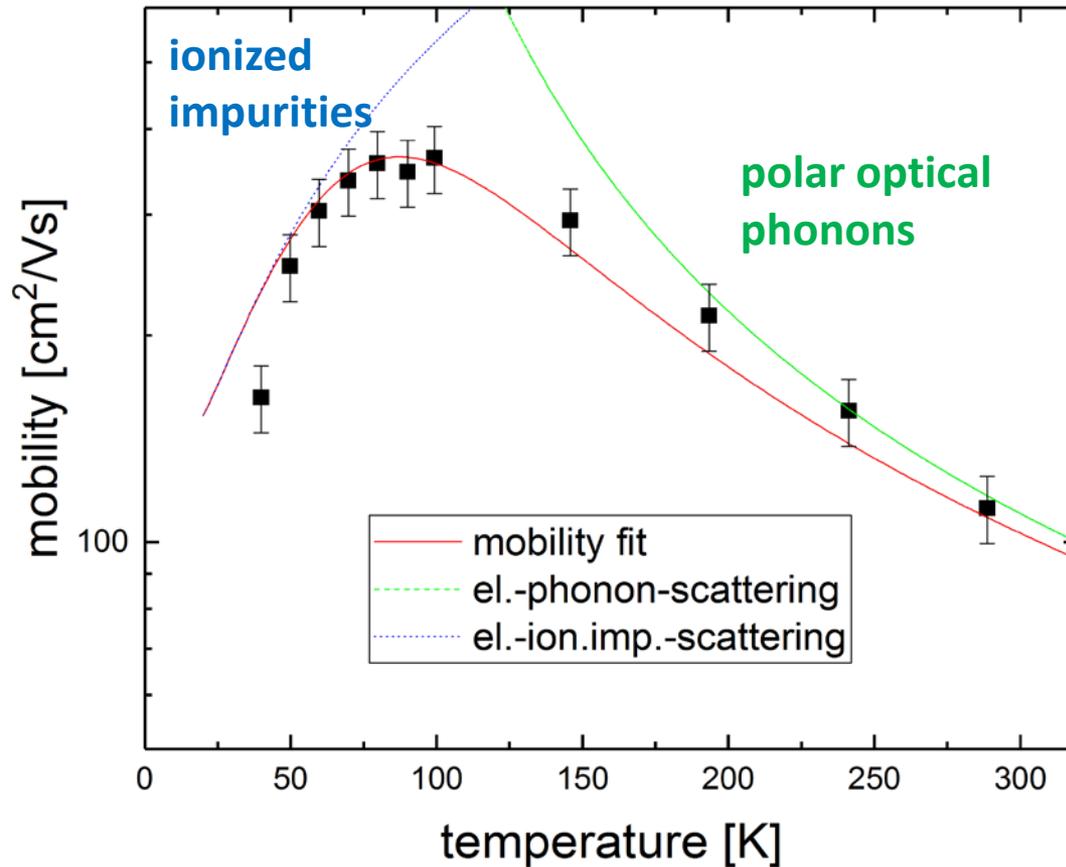
$$\mu = \frac{e\langle\tau_m\rangle}{m^*} = \frac{e}{m^*} \frac{\int_0^\infty E^{3/2} \tau_m(E) f(E) dE}{\int_0^\infty E^{3/2} f(E) dE}$$



scattering  
by  
defects

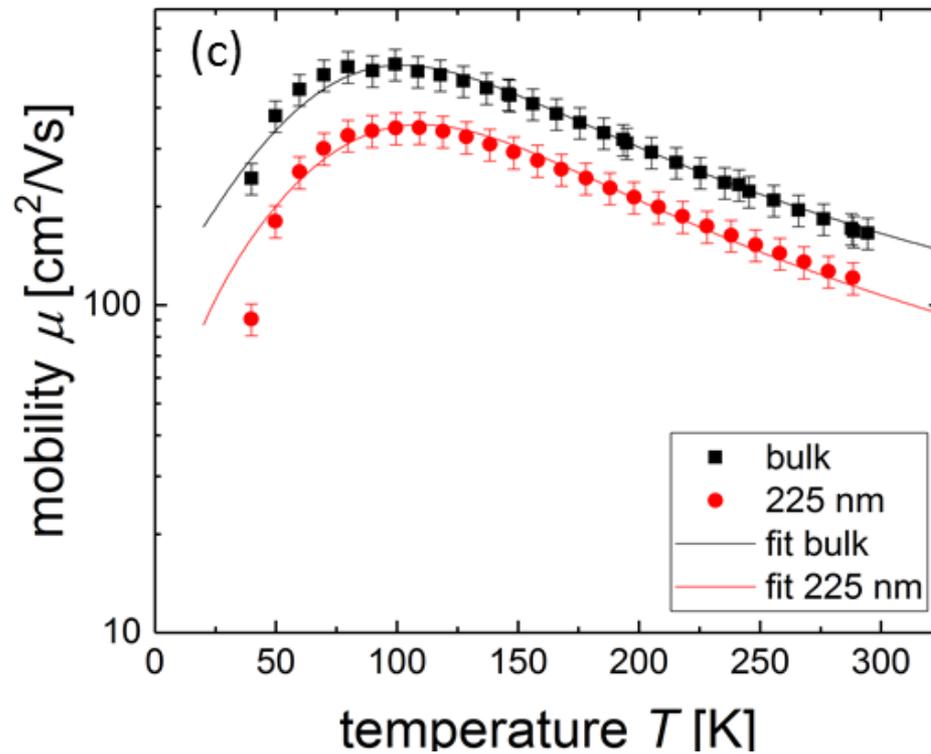
# Charge transport in homoepitaxial $\beta$ -Ga<sub>2</sub>O<sub>3</sub>-bulk

## Dominant scattering mechanisms in $\beta$ -Ga<sub>2</sub>O<sub>3</sub> single crystal



# Charge transport in homoepitaxial $\beta$ -Ga<sub>2</sub>O<sub>3</sub>-films

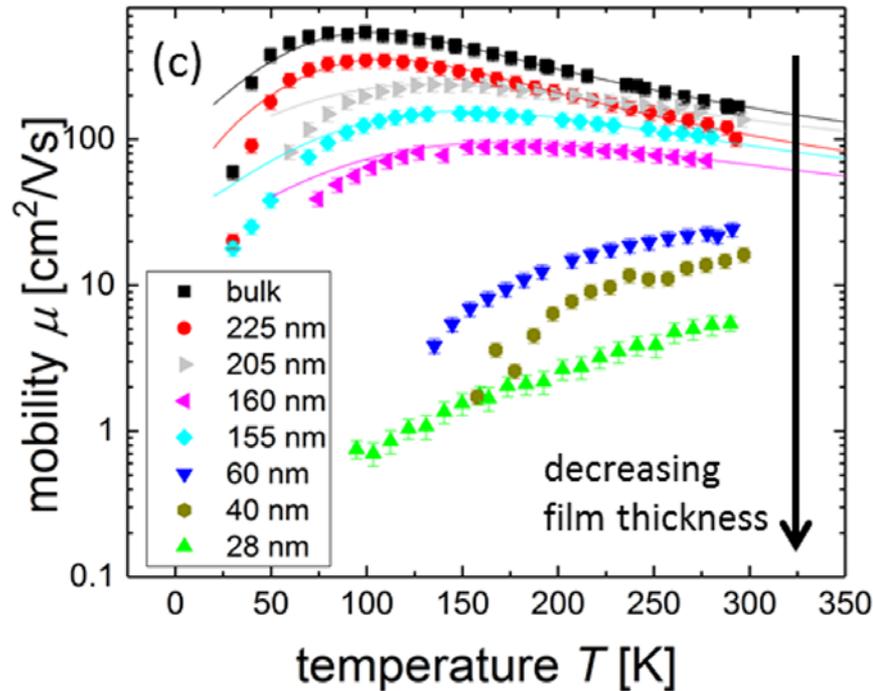
single crystal bulk & 200 nm film



➔ high-quality  
homoepitaxial  
films

# Charge transport in homoepitaxial $\beta$ -Ga<sub>2</sub>O<sub>3</sub>-films

## Mobility suppression with decreasing film thickness



### Real films:

- Neutral impurities, hopping transport
- **Twin boundary scattering**

$$\mu_{\text{tb}} = \frac{eL}{\sqrt{8k_{\text{B}}T\pi m^*}} \exp\left(-\frac{E_{\text{B}}}{k_{\text{B}}T}\right)$$

J.W. Orton, *et al.*;  
Rep. on Prog. in Physics **43**, 1263 (1980).  
R. Schewski, *et al.*;  
J. of Appl. Phys. **120**, 225308 (2016).

- ... but also for **ideal thin films**:

➔ **Surface scattering**  
&  
**boundary effects**

R. Ahrling, *et al.*; Scientific Reports **9**, 13149 (2019).

# Charge transport in homoepitaxial $\beta$ -Ga<sub>2</sub>O<sub>3</sub>-films

## Surface scattering & boundary effects

- length scales:

thickness  $t$   
mean free path  $l$   
de Broglie wavelength  $\lambda_e$   
surface roughness  $r_s$

- In metals:  $l \approx t, \lambda_e \ll l, r_s \gg \lambda_e$

- Fuchs-Sondheimer model:  $l/t$  determines mobility

K. Fuchs, Math. Proc. of the Cambridge Philosophical Society **34**, 100 (1938).  
E. Sondheimer, Advances in Physics **1**, 1 (1952).

- Here:  $\lambda_e \approx t, \lambda_e \gg l, \lambda_e \gg r_s$

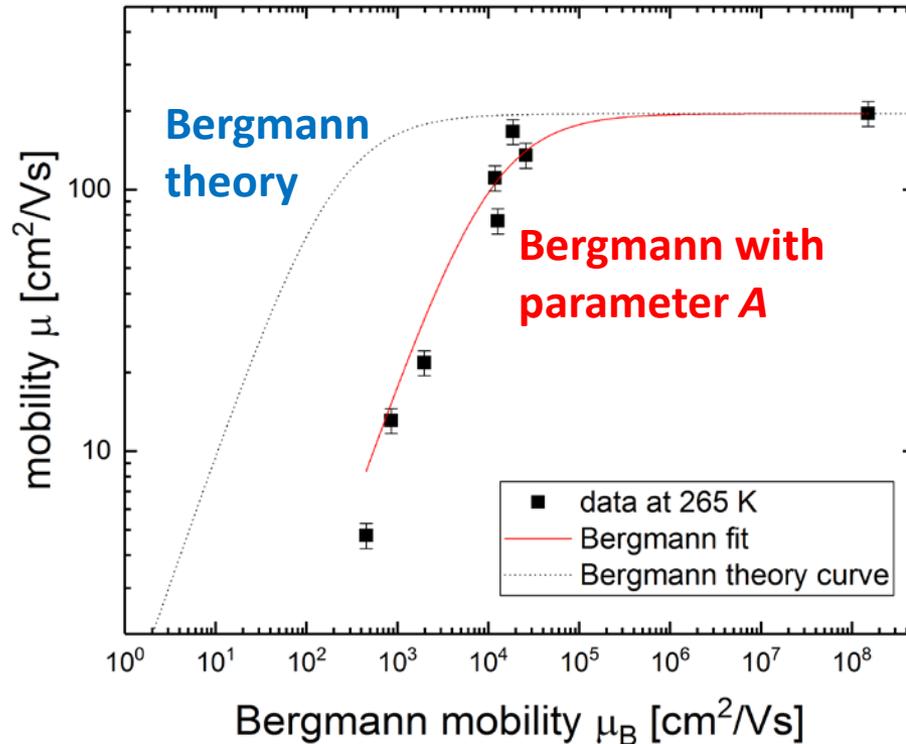
- Bergmann model: quantum mechanical waveguide effect

$$\mu_{\text{Bergmann}} = \frac{e}{\hbar} \left( \frac{t}{\lambda_e} \right)^2 \ln \left( \frac{t}{\lambda_e} \right) \frac{1}{nt}$$

G. Bergmann, *et al.*; PRL **94**, 106801 (2005).

# Charge transport in *ideal* thin films

## Bergman model



$$\mu_{\text{tot}} = \left( \frac{1}{A \cdot \mu_{\text{Bergmann}}} + \frac{1}{\mu_{\text{vol}}} \right)^{-1}$$

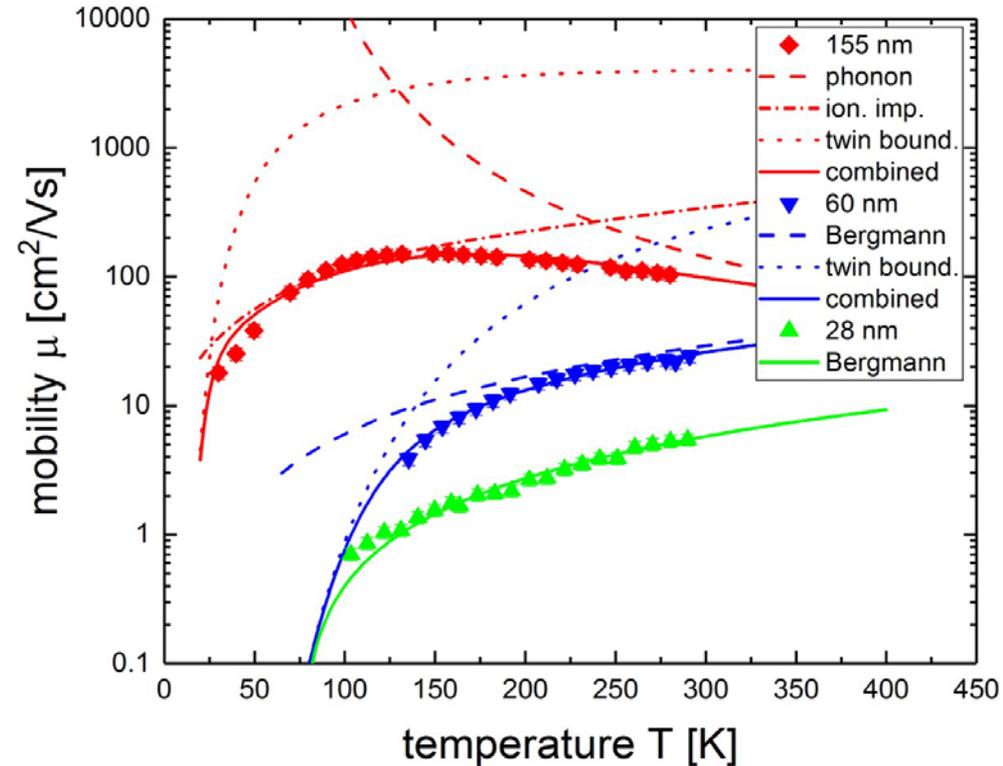
- Mobilities fit to Bergmann model
- Quantitative agreement for  $A = 0.02$

G. Bergmann, *et al.*,  
PRL **94**, 106801 (2005).

# Charge transport in homoepitaxial $\beta$ -Ga<sub>2</sub>O<sub>3</sub>-films

## Summary

- **Thick homoepitaxial (100)  $\beta$ -Ga<sub>2</sub>O<sub>3</sub> films (above 150 nm) behave bulk-like**
  - Optical phonon scattering dominates  $\mu$  for high  $T$
  - Ionized impurity scattering dominates  $\mu$  for low  $T$
- **Thin films (below 100 nm) decrease in and change in  $\mu(T)$  behavior**
  - Additional scattering mechanism occurs  $\rightarrow$  mobility reduction
  - Ideal films: Described by quantum mechanical waveguide effect
- Mobility reduction has to be taken into account for use of thin  $\beta$ -Ga<sub>2</sub>O<sub>3</sub> films in devices



# Thermal properties

## Bulk



Anisotropy in thermal conductivity

## Homoepitaxial films



Phonon-transparent interfaces

Ballistic phonon transport

# Thermal transport measurements

**Thermal conduction differential equation:** 
$$\frac{\partial^2 \Delta T(r, t)}{\partial r^2} + \frac{1}{D} \frac{\partial \Delta T(r, t)}{\partial t} = 0$$

thermal diffusivity:  $\hat{D}$   
thermal conductivity:  $\hat{\lambda} = \hat{D} \cdot C_V \cdot \rho$

D. G. Cahill, *et al.*;  
Phys. Rev. B, **50**, 6077 (1994).

## Experimental setup – electrical line heater



**Solution:**

heating power  $P$  **heater**

Bessel function (zero order second kind)

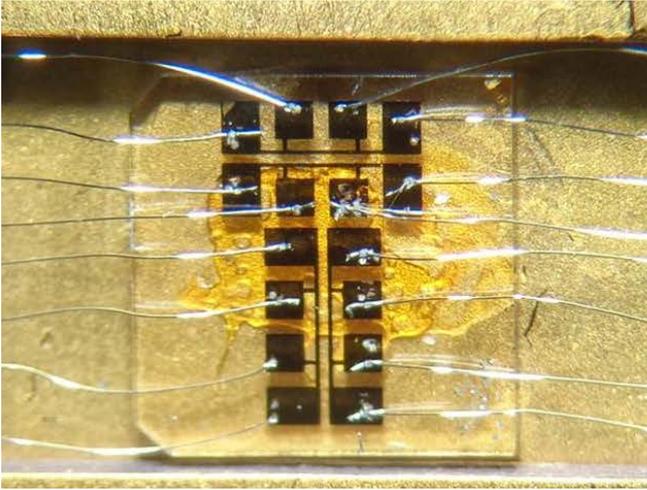
$$\Delta T(r) = \frac{P}{2\pi\hat{\lambda}L} K_0(qr)$$

inverse thermal penetration depth

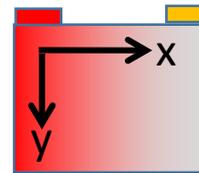
$$q = \sqrt{i2\omega/\hat{D}}$$

# Thermal transport measurements

## 2 $\omega$ -method for anisotropy characterization



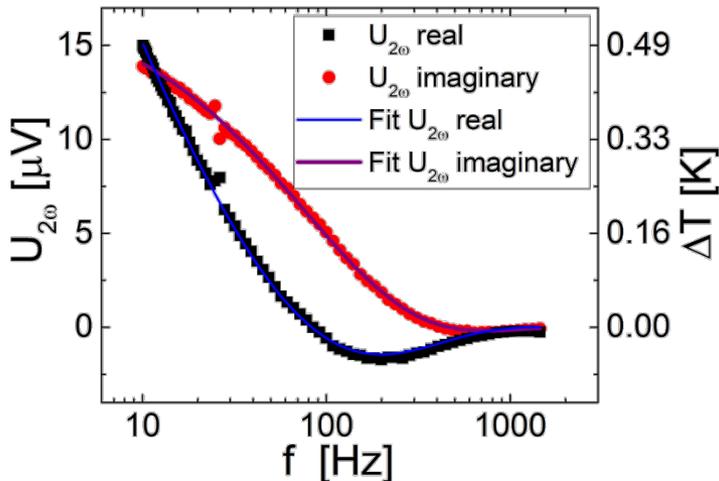
Heater    Sensor



A. T. Ramu and J. E. Bowers;  
*Rev. Sci. Instr.* **83** 124903 (2012).

$$\Delta T = \frac{P}{\pi L \bar{\lambda}} \frac{1}{2\omega_h} \int_{-\omega_h}^{\omega_h} \frac{1}{2\omega_s} \int_{-\omega_s}^{\omega_s} K_0(q \cdot (d + o - p)) \, do \, dp$$

$$\propto U_{2\omega}$$



$$\bar{\lambda} = \sqrt{\lambda_x \cdot \lambda_y}$$

$$q = \sqrt{i2\omega / D_x}$$

We obtain:  $\bar{\lambda}_{[100],[001]}$ ,  $\bar{\lambda}_{[100],[010]}$ ,  $D_{[001]}$ ,  $D_{[010]}$

# Thermal transport in $\beta$ -Ga<sub>2</sub>O<sub>3</sub> single crystals

## Room temperature: thermal diffusivity $D$ and thermal conductivity $\lambda$ for bulk

- [100]-oriented Czochralski grown insulating Mg-doped  $\beta$ -Ga<sub>2</sub>O<sub>3</sub> single-crystal

- Diffusivity  $D$  and conductivity  $\lambda$ :

$$\lambda = D \cdot C_V \cdot \rho$$

axis	L.P.[1]	$D$	$\lambda$	$\lambda_{\text{ex,ref}}$ [2]	$\lambda_{\text{theo,ref}}$ [3]
	Å	mm <sup>2</sup> s <sup>-1</sup>	Wm <sup>-1</sup> K <sup>-1</sup>	Wm <sup>-1</sup> K <sup>-1</sup>	Wm <sup>-1</sup> K <sup>-1</sup>
$a$ [100]	12.2	3.7 ± 0.4	11 ± 1	11 ± 1	16
$b$ [010]	3.0	9.6 ± 0.5	29 ± 2	27 ± 2	22
$c$ [001]	5.8	7.1 ± 0.4	21 ± 2	15 ± 2	21

[1] V. M. Bermudez, *Chem. Phys.* **323** 193 (2006)

[2] Z. Guo *et al.*, *Appl. Phys. Lett.* **106**, 111909 (2015)

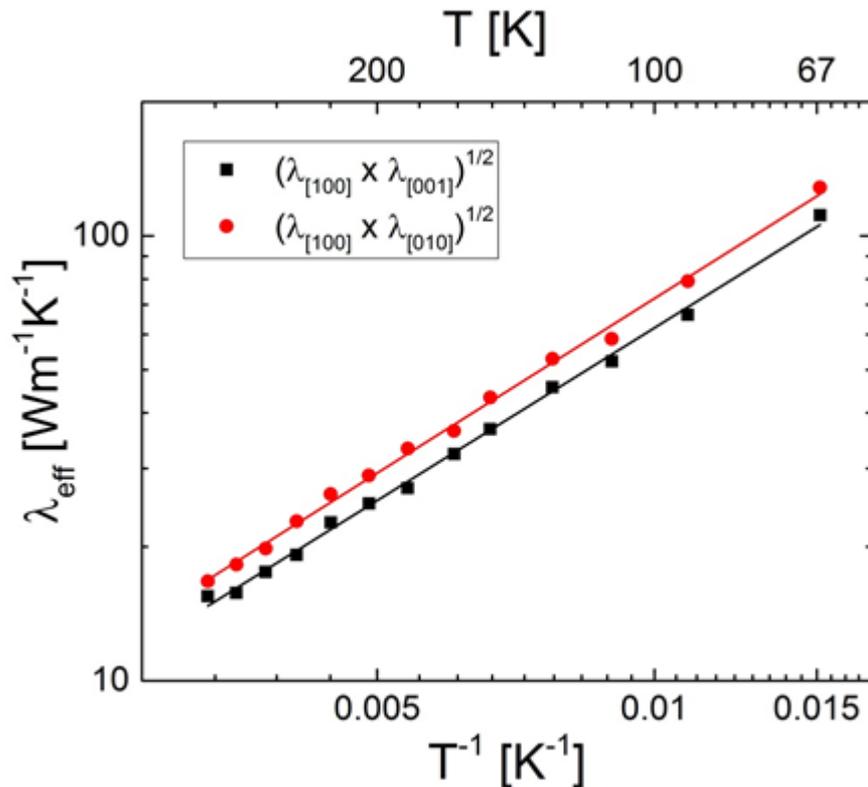
[3] M. D. Santia *et al.* *Appl. Phys. Lett.* **107**, 041907 (2015)

- Highest thermal conductivity value along [010] → (010) no cleavage plane
- Lowest thermal conductivity value along [100] → (100) cleavage plane

# Thermal transport in $\beta$ -Ga<sub>2</sub>O<sub>3</sub> single crystals

## Temperature dependent thermal conductivity $\lambda$

- temperature-independent anisotropy factor:  $\frac{\lambda_{[010]}}{\lambda_{[001]}} = 1.4 \pm 0.1$



M. Handwerg, *et al.*;  
Semicond. Sci. Technol. **31**, 125006 (2016).

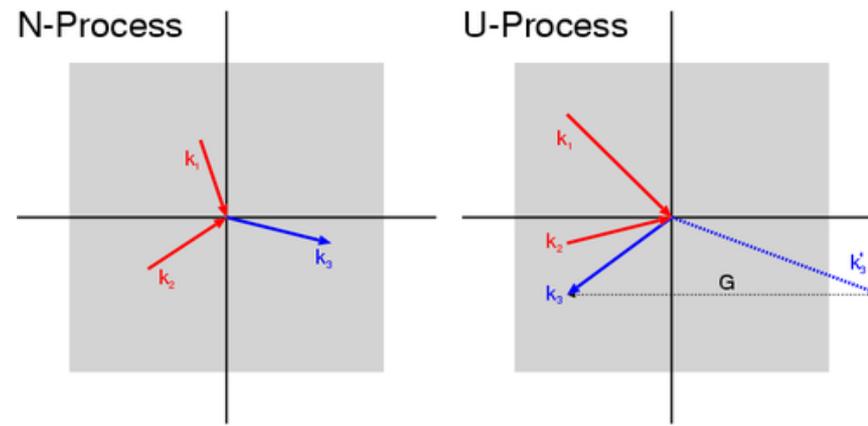
Temperature dependence:

$$\lambda = \frac{1}{3} C_V(T) \Lambda(T) v_s$$

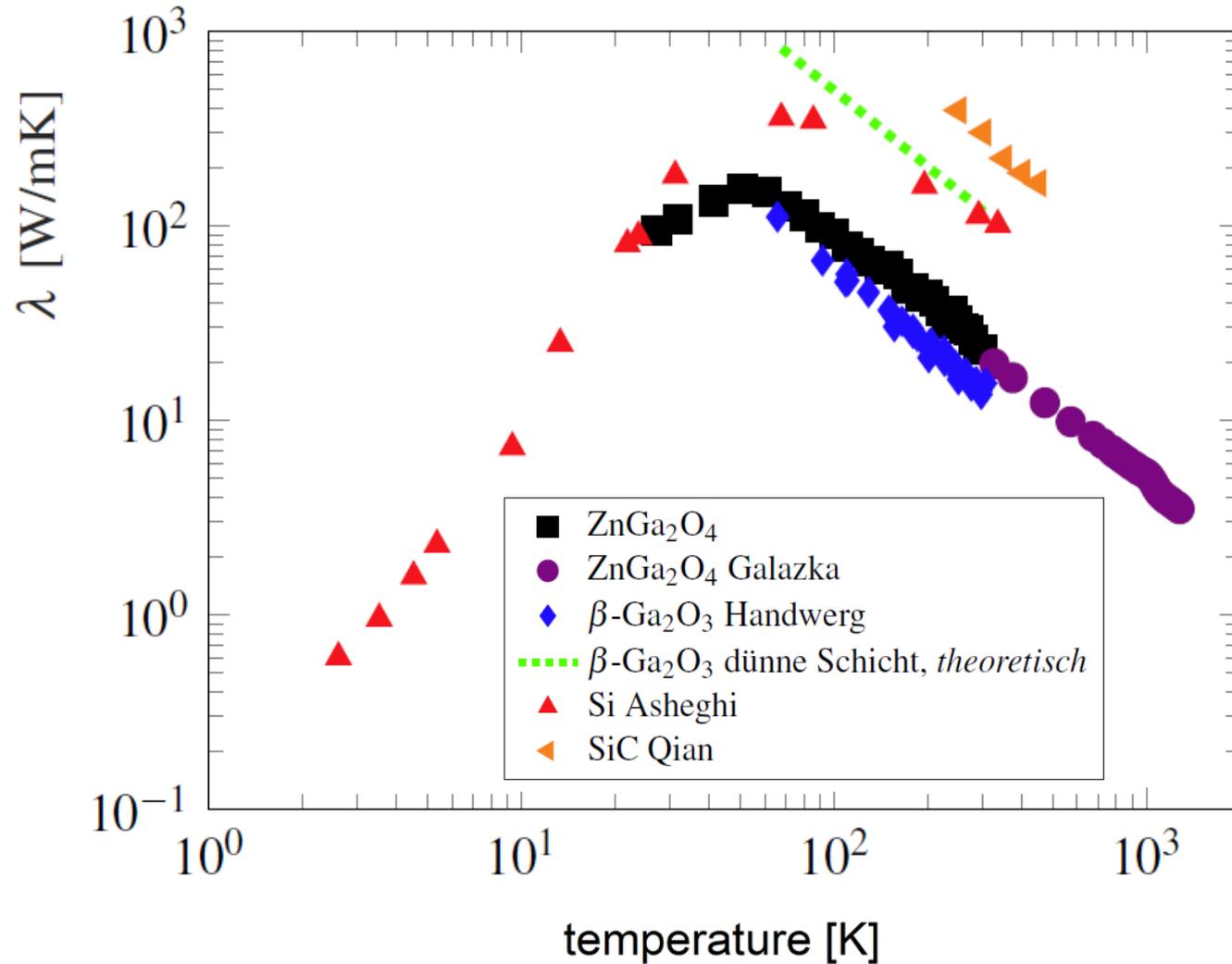
Solid line:

$$C_V \cdot \Lambda \propto T^m \text{ with } m = 1.3 \pm 0.1$$

phonon-phonon-Umklapp-scattering:

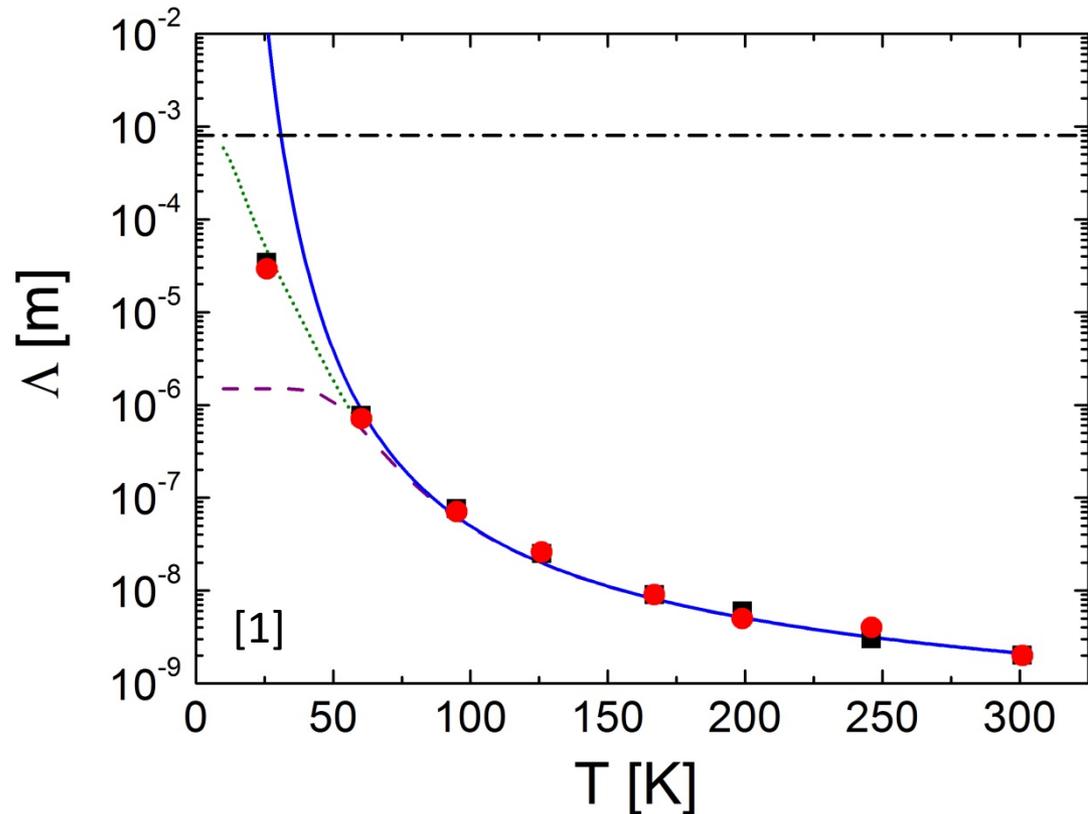


## Comparison: Thermal conductivity $\lambda$



# Thermal transport in $\beta$ -Ga<sub>2</sub>O<sub>3</sub> single crystals

## Phonon mean free path



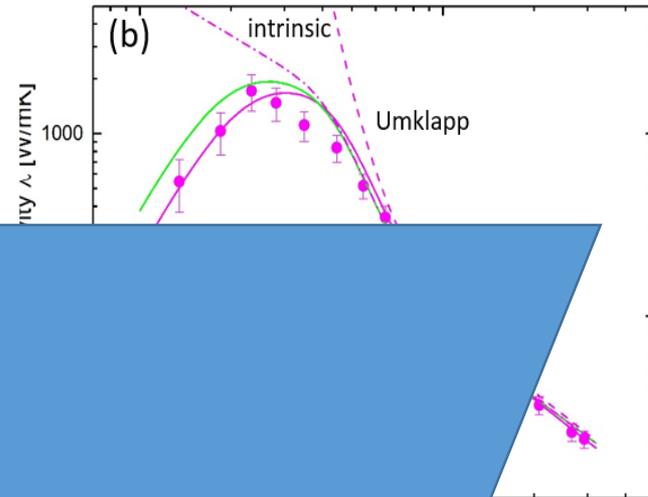
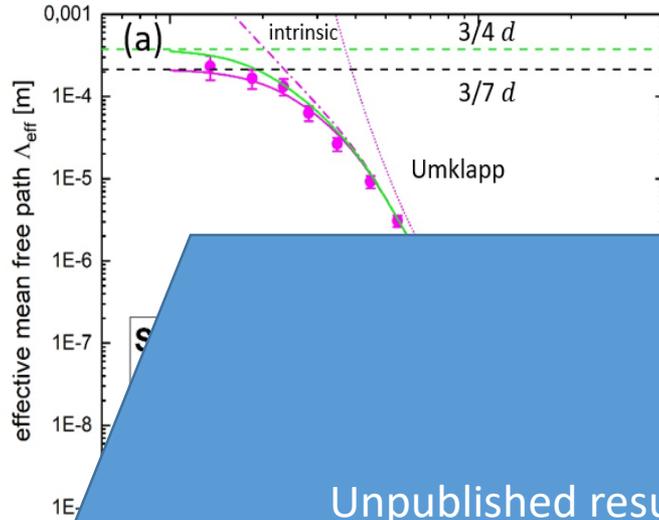
- For low temperatures or thin films the phonon mean free path reaches the sample size:

→ **towards Casimir-limit**

- There the thermal conductivity is maximized

→ **ballistic phonon transport**

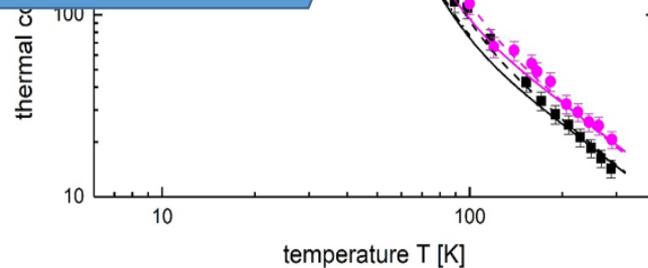
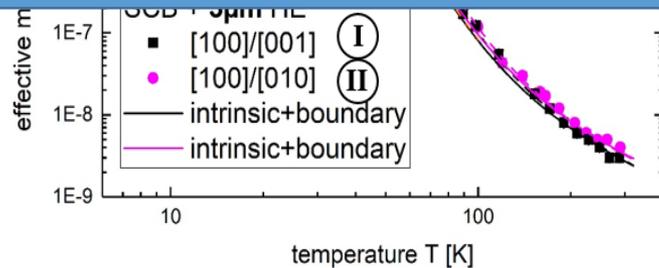
# Thermal transport in $\beta\text{-Ga}_2\text{O}_3$ single crystals + homoepitaxial films



bulk

Unpublished results.  
For pre-/reprint requests:

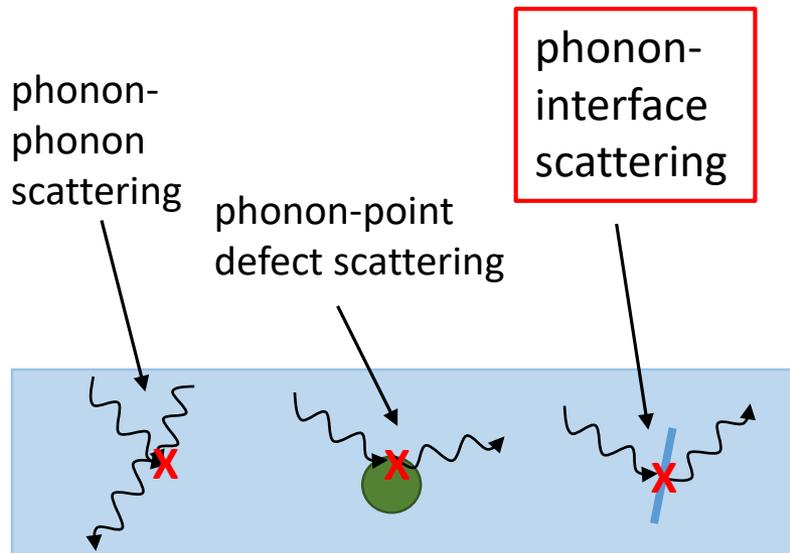
saskia.fischer@hu-berlin.de



bulk  
+  
homoepi  
film

# Phonon scattering mechanisms

phonon 



**Reduced by...**

**Temperature**

**Expitaxy**

**Homoepitaxy**

# Thermal transport in thin films

## Summary

Mg-doped insulating  $\beta\text{-Ga}_2\text{O}_3$  bulk crystals and homo-epi films

- RT:  $\lambda_{[100]} = 11 \pm 1$ ,  $\lambda_{[010]} = 29 \pm 2$  and  $\lambda_{[001]} = 21 \pm 2$  W/(mK)
- Phonon-transparent interface in homoepitaxial films
- Ballistic phonon transport at low temperatures

A remark on polycrystalline films...

- thermal conductivity is decreased due to a reduced phonon mean free path.

M. Handwerg, *et al.*; *Semicond. Sci. Techn.* **30**, 024006 (2015).

M. Handwerg, *et al.*; *Semicond. Sci. Techn.* **31**, 125006 (2016).

R. Mitdank, *et al.*; *Phys. Stat. Sol., A* **211**, 543-549 (2014).

R. Ahrling, PhD Thesis (2023)

# Thermoelectric properties

## Bulk and thick epitaxial films



Diffusive TE & phonon drag

## Thin epitaxial films



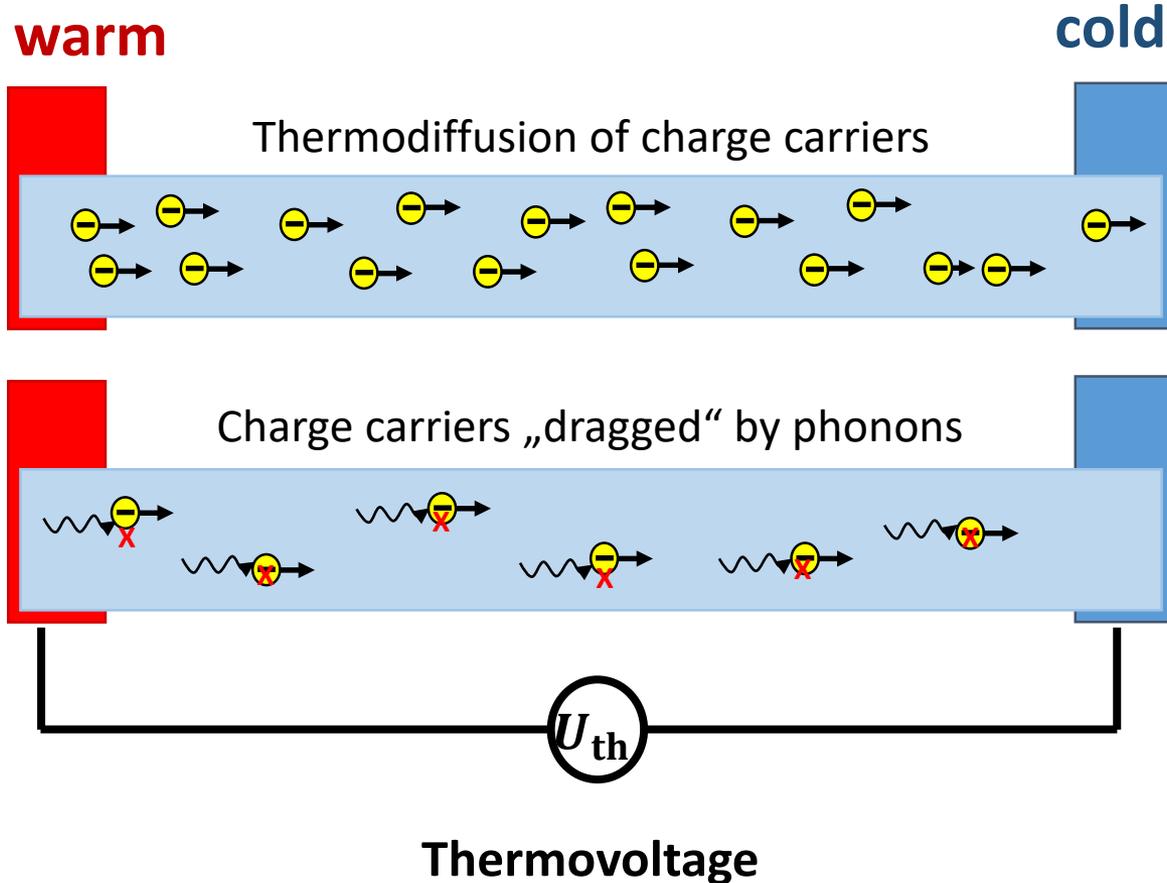
Phonon-transparent interfaces

Giant-phonon drag by design

# Thermoelectric effects in semiconductors

Thermal gradient:  $\Delta T$

Electric field occurs by two processes:



C. Herring,  
Phy. Rev. **96**, 1163 (1954).

Seebeck coefficient:

$$S = -\frac{U_{th}}{\Delta T}$$

# Thermoelectric effects in semiconductors

Seebeck coefficient:

$$S = -\frac{U_{\text{th}}}{\Delta T} = S_{\text{d}} + S_{\text{PD}}$$

thermodiffusion:  $S_{\text{d}} = -\frac{k_{\text{B}}}{e} \left( r + \frac{5}{2} - \eta \right)$

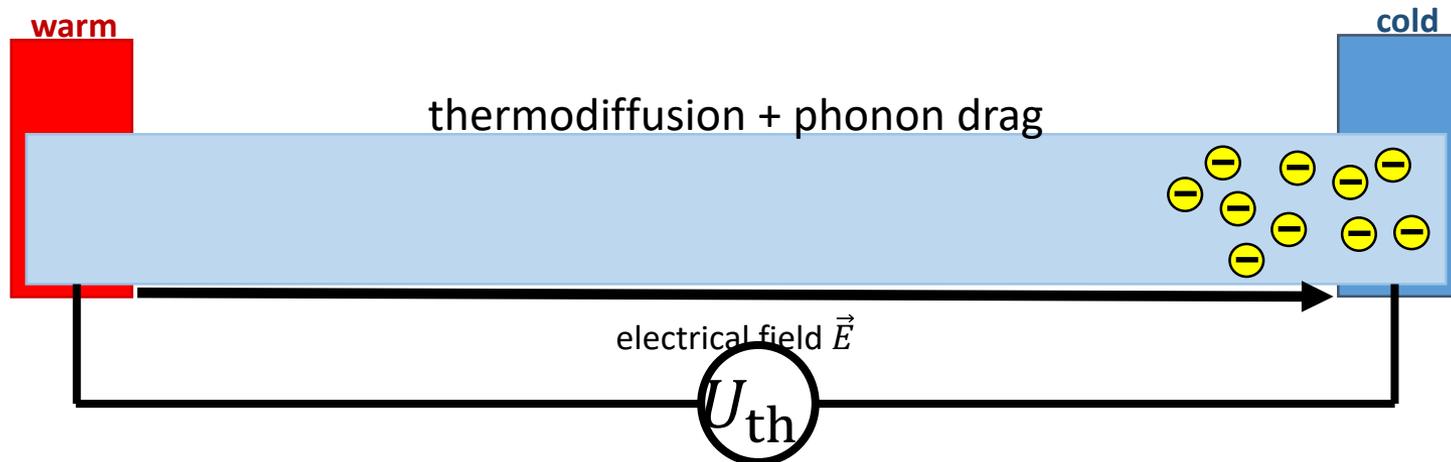
Stratton, Phys. Rev. **126**, 2002 (1962).

phonon drag:  $S_{\text{PD}} = -\frac{v^2}{T} \cdot \frac{1}{\mu_{\text{AP}}} \cdot \tau_{\text{Ph}}$

Herring, Phy. Rev. **96**, 1163 (1954).

Hutson, JAP **32**, 2287 (1961).

Smith and Butcher, J. Physics: Cond. Mat. **2**, 2375–2382 (1990).



# Phonon drag contribution to the Seebeck coefficient

phonon drag:

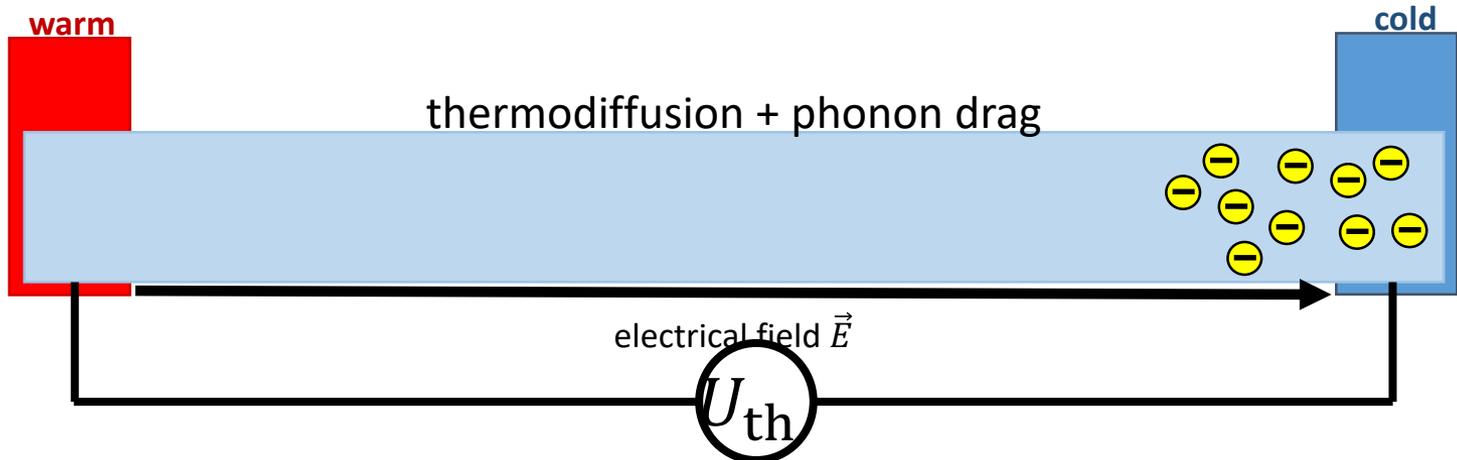
$$S_{\text{PD}} = -\frac{v^2}{T} \cdot \frac{1}{\mu_{\text{AP}}} \cdot \tau_{\text{Ph.}}$$

$$= \frac{m^* v^2}{eT} \cdot \frac{\tau_{\text{Ph.-Ph.}}}{\tau_{\text{El.-Ph.}}}$$

Herring, *Phy. Rev.* **96**, 1163 (1954).

Hutson, *JAP* **32**, 2287 (1961).

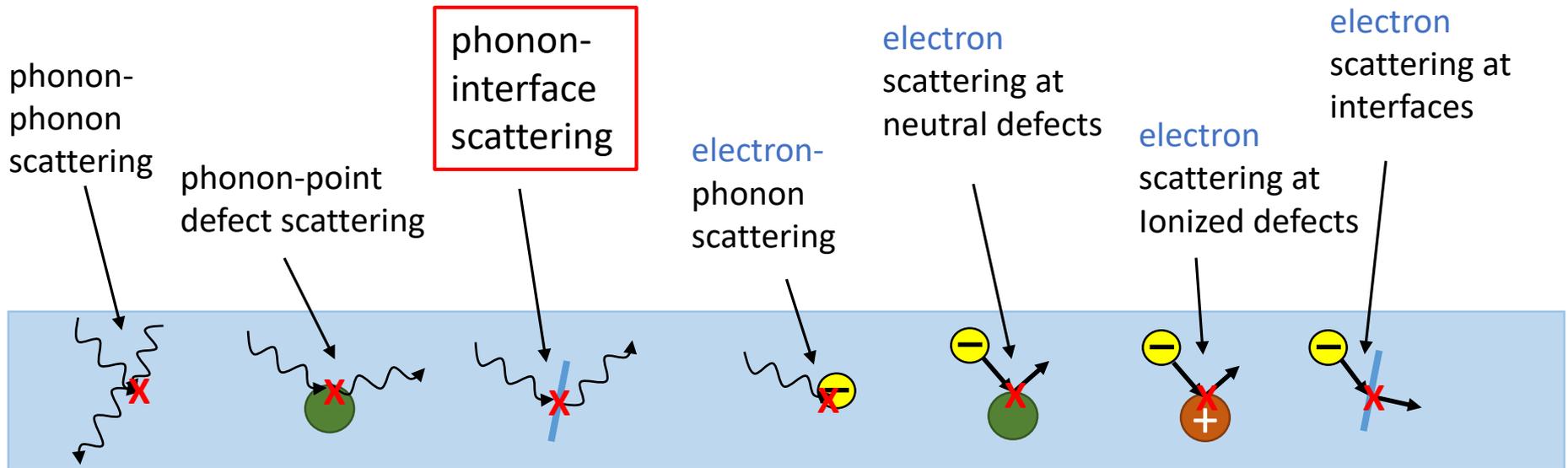
Smith and Butcher, *J. Physics: Cond. Mat.* **2**, 2375–2382 (1990).



# Thermoelectricity: Full „zoo" of scattering mechanisms

phonon 

electron 



Reduced by...

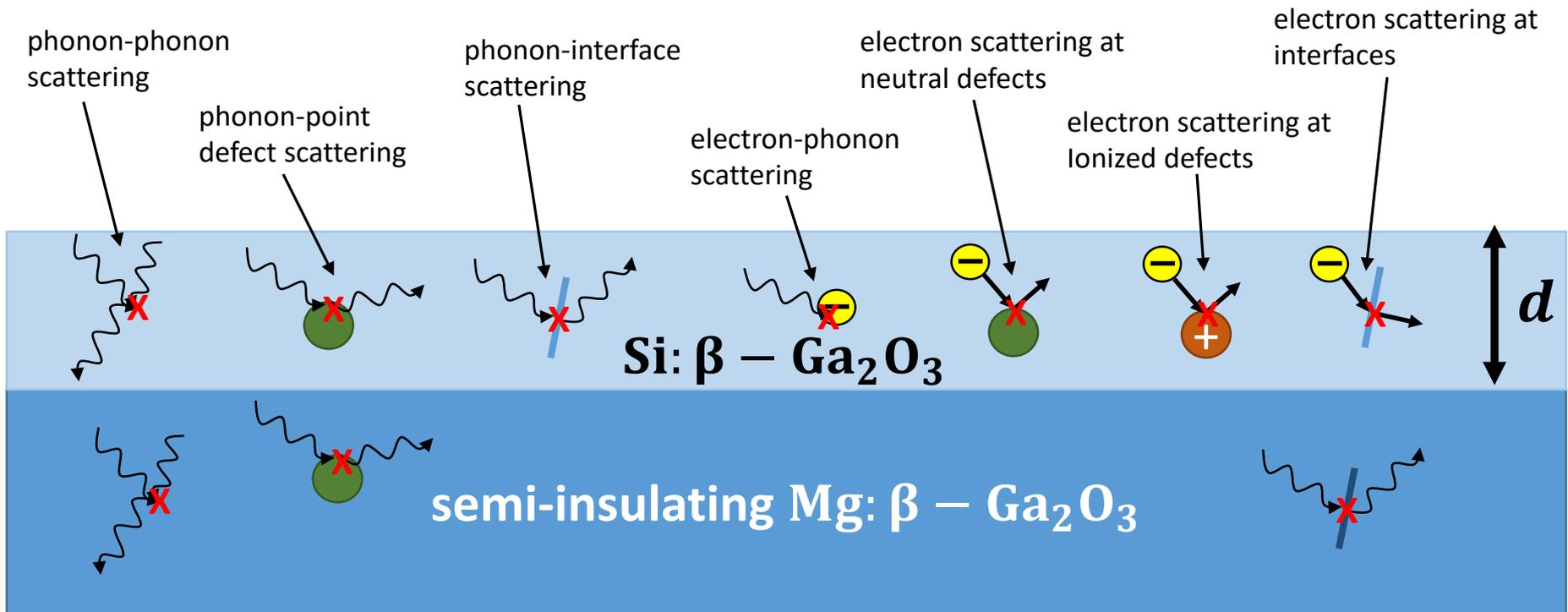
Temperature

Expitaxy

Homoepitaxy

# Homoepitaxial films of $\beta$ -Ga<sub>2</sub>O<sub>3</sub>

→ Selection of in-plane phonons for phonon-drag effects by choosing a film thickness below the phonon mean-free path

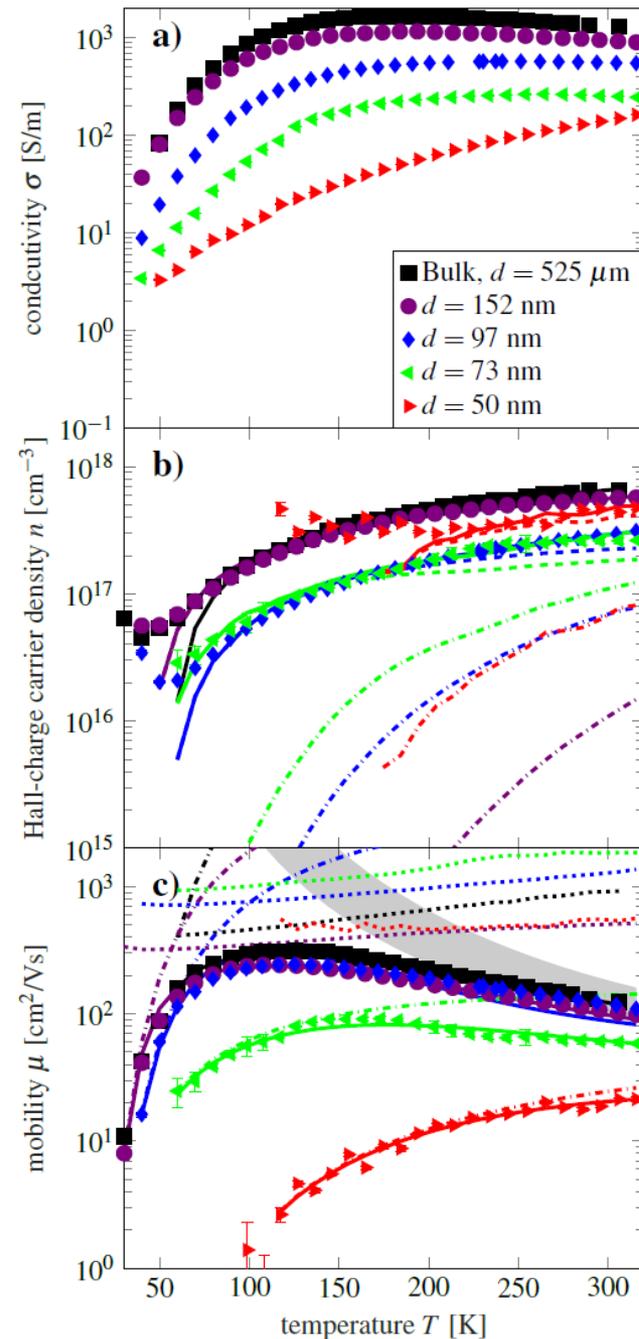


# Homoepitaxial films of $\beta$ -Ga<sub>2</sub>O<sub>3</sub>:

Electrical conductivity

Charge carrier density

Mobility



J. Boy, et al., *APL Mater.* **7**, 022526 (2019).

J. Boy, *PhD Thesis* (2022)

# Homoepitaxial films of $\beta$ -Ga<sub>2</sub>O<sub>3</sub>:

reduced  
chemical potential  $\eta$

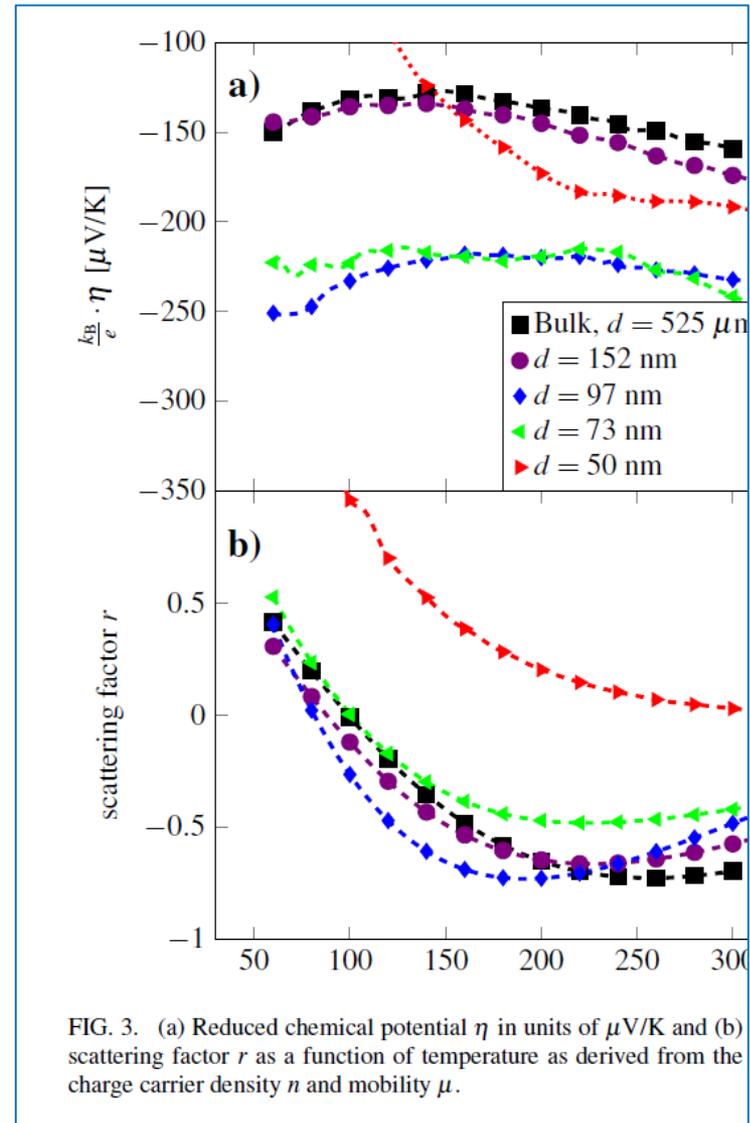
scattering factor  $r$



thermodiffusion:

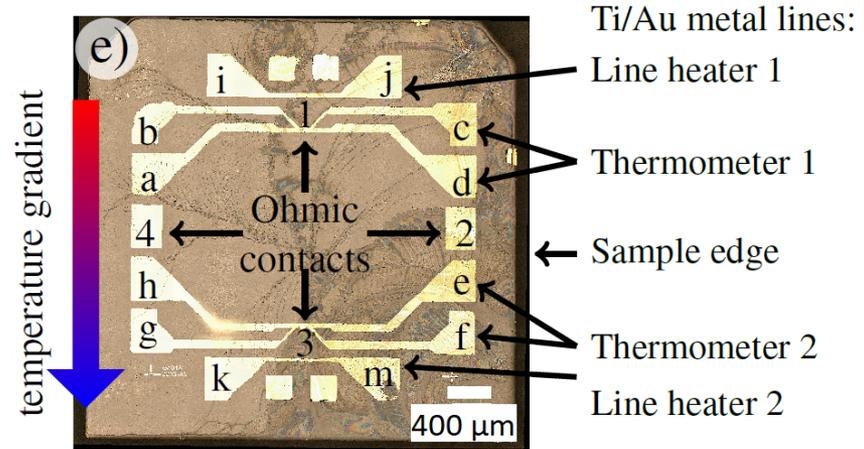
$$S_d = -\frac{k_B}{e} \left( r + \frac{5}{2} - \eta \right)$$

Stratton, Phys. Rev. **126**, 2002 (1962).

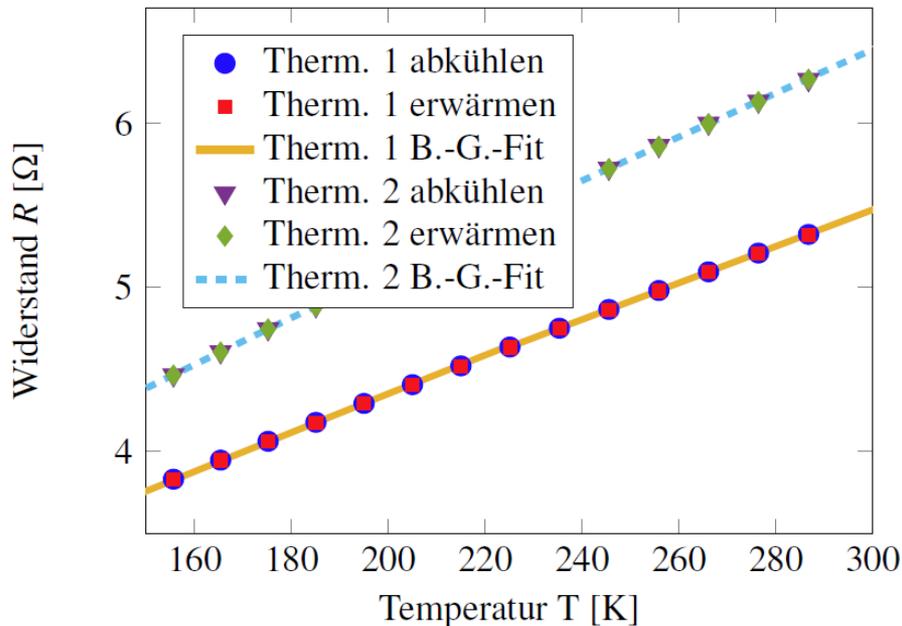


J. Boy, et al., *APL Mater.* **7**, 022526 (2019).

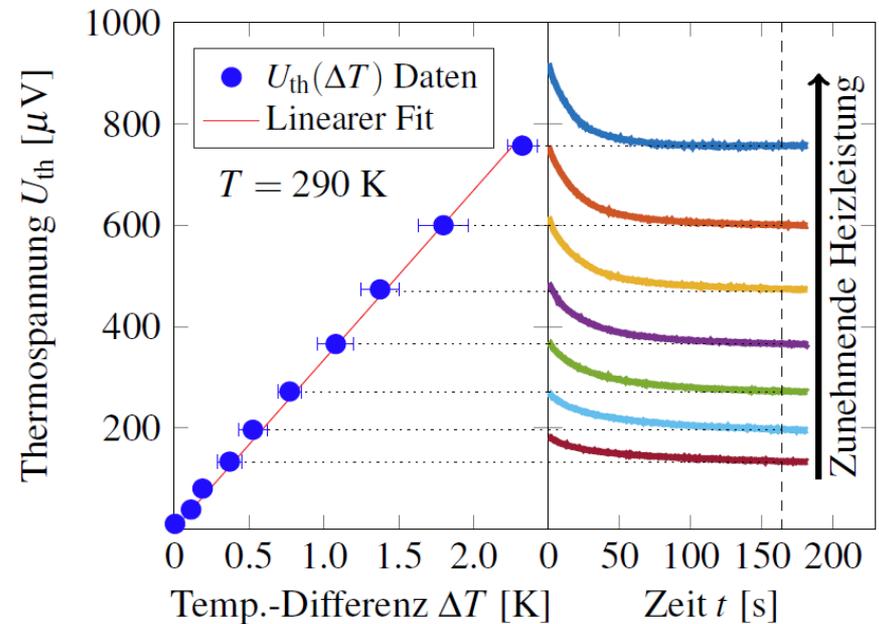
# Thermoelectric micro measurement platform



## Heater lines



## Thermovoltages



# Giant phonon drag increase in $\beta - \text{Ga}_2\text{O}_3$ homoepitaxial thin films

$$S_{\text{PD}} = -\frac{v^2}{T} \frac{1}{\mu_{\text{AP}}} \tau_{\text{Ph.}} = \frac{m^* v^2}{eT} \cdot \frac{\tau_{\text{Ph.}}}{\tau_{\text{El.-Ph.}}}$$

Calc. thermodiffusion part fits well to theoretical work

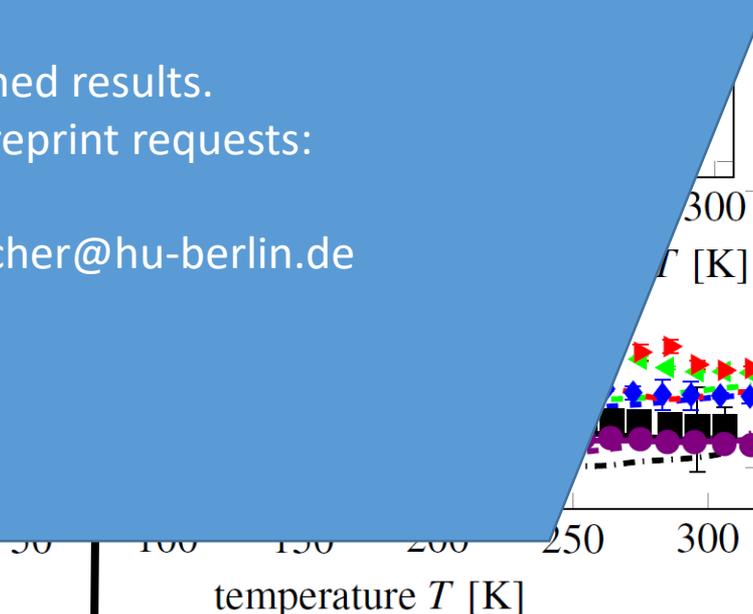
Kumar and Singiseti, *APL* **117**, 262104 (2020).

Strong  
incr  
of  
p

Unpublished results.  
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$S_{\text{PD}}$



$$A = T \cdot S_{\text{PD}} \propto \frac{\tau_{\text{Ph.}}}{\tau_{\text{El.-Ph.}}}$$

thermodiffusion

# Summary - Thermoelectric properties

## Phonon-drag: A measure of electron-phonon interactions

- Thermoelectric voltages and Seebeck-coefficients

## Phonon-transparent interfaces

- Thin film growth by homoepitaxy

## Control of the effective electron-phonon interaction cross-section

- Film thickness below phonon mean-free path



## Giant-phonon drag increase by design

- selection of relevant in-plane phonons

**Outlook:** Results are generally valid for a wide range of materials.-

# Transport properties of $\beta$ – Ga<sub>2</sub>O<sub>3</sub> single crystals and thin films

## Anisotropic thermal conductivity & ballistic phonon transport

Handwerg, *et al.*, *Semicond. Sci. Technol.* 30, 024006 (2015).

Handwerg, *et al.*, *Semicond. Sci. Technol.* **31**, 125006 (2016).

R. Ahrling, *PhD Thesis* (2023)

## Electrical properties & size effects of homoepitaxial thin films & flakes:

R. Mitdank, *et al.*, *Phys. Stat. Sol., A* 211, 543-549 (2014).

R. Ahrling, *et al.*, *Scientific Reports* **9**, 13149 (2019).

## Seebeck coefficients & (Giant-)phonon drag increase by thin film design

J. Boy, *et al.*, *APL Mater.* **7**, 022526 (2019).

J. Boy, *PhD Thesis* (2022)

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